

CHEMICAL SCIENCES

PHYSICO-CHEMICAL INVESTIGATION OF PHASE EQUILIBRIUM AND GLASS FORMATION IN THE AsS-Cu₂Cr₄Te₇ SYSTEM

Ismailova S.

Postgraduate research fellow

Aliyev I.

Doctor of Chemistry, prof., head. lab.

Institute of Catalysis and Inorganic Chemistry, named after M. Nagiyev, Ministry of Science and Education of the Republic of Azerbaijan.

Ismailov T.

Candidate of Medical Sciences, associate professor.

Azerbaijan State Medical University named after N. Narimanov, Baku.

Ahmedova C.

Doctor of Chemistry, prof.,

Adiyaman University, Faculty of Arts and Sciences, Department of Chemistry, Turkiye

<https://doi.org/10.5281/zenodo.17226043>

Abstract

Phase formation and vitrification in the AsS-Cu₂Cr₄Te₇ system are studied by the methods of physical and chemical analysis: differential thermal (DTA), X-ray diffraction (XRD), microstructural (MKA), and also determination of microhardness and density. The T-x phase diagram is constructed. It is established that the AsS-Cu₂Cr₄Te₇ system is a partially quasibinary section. Based on the results of the microstructural analysis, it was established that in the AsS-Cu₂Cr₄Te₇ system at room temperature, the solubility of 12 mol.% AsS based on the Cu₂Cr₄Te₇ compound was found, while the area of the solid solution based on the AsS compound is practically undefined. During slow cooling in the AsS-Cu₂Cr₄Te₇ system, a region of glass formed on the basis of AsS 10 mol. % Cu₂Cr₄Te₇. The area of AsS-based glass when cooled in ice water is 15 mol.% Cu₂Cr₄Te₇. In the interval of concentrations of 10–30 mol. % Cu₂Cr₄Te₇ there are alloys of glass crystal composition.

Keywords: peritectic, quasibinary, glass, solid solution, microhardness.

Introduction

Arsenic chalcogenides, one of the main elements of Group V, are known to form in a glassy state. Recently, arsenic chalcogenides, as well as complex glassy alloys and compounds based on them, have attracted the attention of researchers due to their unique functional properties. Arsenic chalcogenides, as well as complex glassy alloys and compounds based on them, are widely used in various fields of photoelectronics as photosensitive, luminescent, and acousto-optic materials [1–8]. Chromium chalcogenides form magnetic materials with a number of chalcogenides, including copper chalcogenides [9–15].

We have studied a number of systems involving the compound Cu₂Cr₄Te₇ [16, 17]. The physical properties of arsenic monosulfide AsS remain virtually unexplored. Therefore, it can be expected that new complex phases and glassy alloys obtained by the chemical interaction of AsS and Cu₂Cr₄Te₇ chalcogenides can also be semiconductor materials with magnetic and magneto-optical properties.

The aim of this work is to conduct a physicochemical study of the phase equilibrium in the AsS-Cu₂Cr₄Te₇ system, construct its phase diagram, and determine the glass transition and solid solution boundaries.

The glassy compound AsS melts at 318°C with an open maximum and crystallizes in the orthorhombic syngony, with lattice parameters: $a = 9.32$; $b = 13.546$;

$c = 6.585 \text{ \AA}$ [18]. The compound Cu₂Cr₄Te₇ melts incongruently at 1000°C and has a wide homogeneity region [15].

Experimental Section

The synthesis of AsS-Cu₂Cr₄Te₇ alloys was carried out in two stages. In the first stage, AsS and Cu₂Cr₄Te₇ compounds were synthesized using the ampoule method. The AsS compound was obtained from the elements in a single-zone furnace at a temperature of 400–700°C. The AsS compound was obtained as a glass with normal cooling. A two-zone furnace was used to obtain the Cu₂Cr₄Te₇ compound from the elements. To synthesize the Cu₂Cr₄Te₇ compound of the AsS-Cu₂Cr₄Te₇ system, elements taken in stoichiometric composition are placed in a 20-cm-long ampoule. To synthesize the Cu₂Cr₄Te₇ compound, part of the ampoule is inserted into a furnace at 1100°C. The remaining portion of the ampoule is left outside the furnace. During this process, Te vapor initially begins to form. This process continues until the Te vapor disappears. After maintaining the reaction at 1100°C for 3 hours, the furnace temperature is increased to 950°C, held for 4 hours, and then cooled by disconnecting it from the power source. In the second stage, the AsS-Cu₂Cr₄Te₇ alloys were synthesized from the AsS and Cu₂Cr₄Te₇ compounds using the ampoule method. The alloys of the system are then analyzed using physicochemical methods.

Differential-thermal analysis was performed on a Thermoscan-2 pyrometer. Chromel-alumel was used as a thermocouple, and the heating rate was 5°C/min.

Microstructural analysis was performed on an MIM-8 microscope. $\text{HNO}_3 + \text{HF}$ solutions (1:2) were used as etchants to reveal phase boundaries in the samples. X-ray phase analysis of the alloys was performed on a D2 PHASER X-ray diffractometer. $\text{CuK}\alpha$ radiation and a Ni filter were used as the anode.

Microhardness was measured on a PMT-3 metallographic microscope. Density was determined pycnometrically, using toluene as the filler solution.

Results and Discussion

Alloys of the $\text{AsS-Cu}_2\text{Cr}_4\text{Te}_7$ system are compact, and their color ranges from red to black. The alloys of the system are resistant to air, water, and organic solvents. They are highly soluble in strong acids (HNO_3 , H_2SO_4). Alloys rich in AsS are soluble in alkalis (NaOH , KOH).

It should be noted that the physicochemical analyses of the alloys were carried out in both the crystalline and glassy states. Thermal analysis of the alloys before heat treatment revealed that thermal effects were

observed in the thermograms of samples in the range of 0–15 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$ at a temperature of 170°C , which corresponds to the softening temperature of glass. The composition of the alloys of the system was studied by microstructural analysis, and it was found that there is a wide range of solubility in the region of the $\text{Cu}_2\text{Cr}_4\text{Te}_7$ compound. The microstructures of the alloys in the range of 0–15 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$ appear as opaque single-phase. Microstructural analysis revealed that only single-phase alloys were obtained based on the $\text{Cu}_2\text{Cr}_4\text{Te}_7$ compound. Figure 1 shows the microstructure of alloys with concentrations of 40, 70, and 90 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$. As can be seen from the figure, alloys with concentrations of 40 and 70 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$ are two-phase. The alloy with a concentration of 90 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$ is single-phase and represents a solid solution based on the $\text{Cu}_2\text{Cr}_4\text{Te}_7$ compound.



Fig. 1. Microstructure of $\text{AsS-Cu}_2\text{Cr}_4\text{Te}_7$ alloys.
1–40 mol. %, 2–70 mol. %, 4–90 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$.

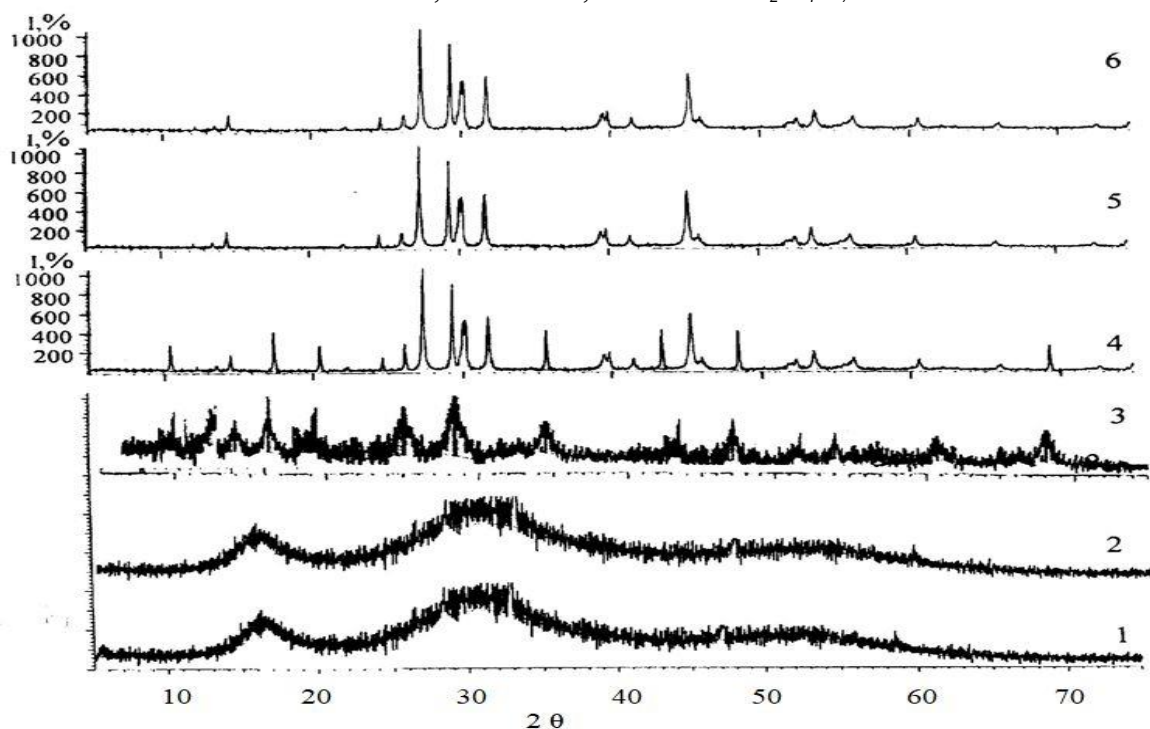


Fig. 2. X-ray diffraction patterns of $\text{AsS-Cu}_2\text{Cr}_4\text{Te}_7$ alloys.
1–5, 2–10, 3–30, 4–50, 5–90, 6–100 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$.

To clarify the results of differential thermal and microstructural analyses of $\text{AsS-Cu}_2\text{Cr}_4\text{Te}_7$ alloys, X-ray diffraction analysis was performed on alloys with contents of 5, 10, 30, 50, and 90 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$ (Fig.

2). Diffraction maxima in the diffraction patterns of alloys in the range of 0–30 mol. % $\text{Cu}_2\text{Cr}_4\text{Te}_7$ in the studied system revealed weak lines.

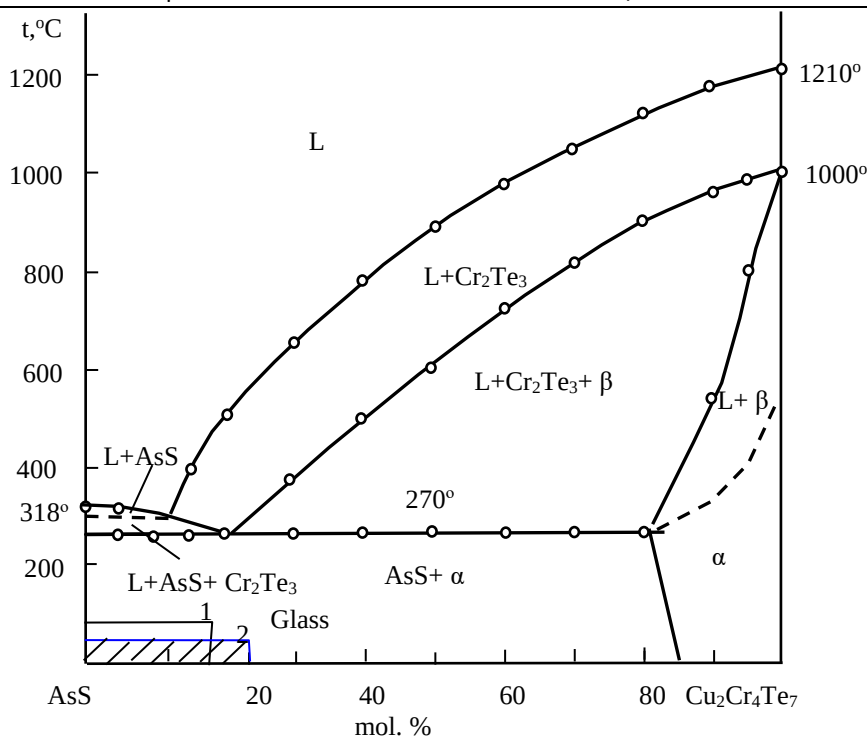


Fig.3. *T-x* phase diagram of the AsS-Cu₂Cr₄Te₇ system.
1- during slow cooling, 2- during cooling in ice water, glass area

Table 1.

Physico-chemical properties of glassy alloys of the AsS-Cu₂Cr₄Te₇ system (glass region)

Composition, mol. %		Thermal effects, °C			Microhardness, MPa	Density, q/cm ³	MSA
AsS	Cu ₂ Cr ₄ Te ₇	T _g	T _{crus.}	T _{melt.}			
100	0	170	220	318	1350	3,35	Șüşə
97	3	170	220	315	1380	3,55	—
95	5	175	230	315	1380	3,60	—
93	7	180	235	320	1380	3,67	—
90	10	190	240	270	1385	3,72	—
85	15	190	245	400	1390	3,90	Glass-crust
80	20	190	250	510	1400	4,10	Glass-crust
70	30	190	265	380	1430	4,47	Glass-crust
60	40	190	270	450	1440	4,85	crustal

This indicates the glassy nature of the samples in this region. Diffraction maxima were observed in the X-ray diffraction patterns of alloys containing 50 and 90 mol.% Cu₂Cr₄Te₇. Analysis revealed that the diffraction lines in the diffraction pattern of the sample containing 90 mol. % Cu₂Cr₄Te₇ are identical to those of the Cu₂Cr₄Te₇ compound, indicating that this sample is a solid solution alloy based on Cu₂Cr₄Te₇.

Based on the results of a comprehensive physico-chemical analysis, a phase diagram of the AsS-Cu₂Cr₄Te₇ system was constructed (Fig. 3). In the AsS-

Cu₂Cr₄Te₇ system, glassy AsS-based alloys with a composition of 10 mol. % Cu₂Cr₄Te₇ formed upon slow cooling. Alloys with a glassy-crystalline composition are present in the layer range of 10–30 mol. % Cu₂Cr₄Te₇. Table 1 shows the compositional dependences of the softening and crystallization temperatures, microhardness, and density of alloys from the glassy and glassy-crystalline regions. Table 2 presents some physicochemical properties of the alloys in the crystalline state of the system.

Table 2.

Composition, results of differential thermal analysis, determination of microhardness and density of AsS-Cu₂Cr₄Te₇ system alloys in crystalline form

Composition, mol. %		Thermal effects, °C	Density, q/cm³	Microhardness, MPa	
AsS	Cu₂Cr₄Te₇			AsS	α
				0,15 H	0,20 H
100	0,0	318	3,52	660	-
95	5,0	270,315	3,60	700	-
90	10	270	3,75	720	-
85	15	270,400	4,05	740	-
80	20	270,510	4,24	-	-
70	30	270,380,650	4,59	-	-
60	40	270,450,780	4,95	-	2000
50	50	270,600,890	5,33	-	2000
40	60	270,720,975	5,67	-	2000
30	70	270, 810,1050	6,03	-	2000
20	80	270,900,1120	6,38	-	2000
10	90	545,960,1175	6,74	-	2000
5,0	95	800,980	7,22	-	2000
0.0	100	1000,1210	7,10	-	1970

The liquidus of the AsS-Cu₂Cr₄Te₇ system represents the initial crystallization curves of the β -solid solution obtained from Cu₂Cr₄Te₇. In the system, primary crystals of the AsS compound precipitate in the range of 0–20 mol. % Cu₂Cr₄Te₇. The system undergoes a process of eutectic equilibrium and peritectic transformation.

The decomposition of the Cu₂Cr₄Te₇ compound results in the formation of three-phase alloys consisting of (L + AsS + Cr₂Te₃) and (L + Cr₂Te₃ + β). Since the peritectic process $L + Cr_2Te_3 \leftrightarrow Cu_2Cr_4Te_7$ occurs, two-phase alloys consisting of AsS + α crystallize below the solidus line in the system.

The microhardness and density of AsS-Cu₂Cr₄Te₇ alloys were studied as a function of their composition in the glassy and crystalline states (Tables 1 and 2).

Two different microhardness values were determined in the system. Before heat treatment, i.e. in the glassy state, the microhardness and density of glasses obtained on the basis of AsS are $H_\mu = (1350-1440)$ MPa, $\rho = (3.35-4.85)$ g/cm³, for α -solid solutions based on the compound Cu₂Cr₄Te₇ the microhardness is $H_\mu = (2750-2800)$ MPa, and the density $\rho = 3.52-7.22$ g/cm³. After heat treatment, the microhardness of samples from the glassy region is $H_\mu = (660-740)$ MPa, $\rho = (3.53-4.59)$ g/cm³. The microhardness and density of the crystalline phases do not change. It was found that during the glass-crystal transition the hardness of the alloys decreases and their specific gravity increases.

Conclusion

Thus, based on the results of comprehensive physicochemical analysis methods, a phase diagram of the AsS-Cu₂Cr₄Te₇ system has been constructed. The phase diagram of the system is partially quasi-binary and belongs to the eutectic type. The system undergoes eutectic equilibrium and peritectic transformation, leading to the crystallization of two-phase alloys of the AsS + α composition below the solidus line.

In the AsS-Cu₂Cr₄Te₇ system, the AsS-based solid solution region is virtually undefined. The Cu₂Cr₄Te₇-

based solid solution region is 15 mol. % AsS. The glass transition region of the AsS-based system under normal conditions is determined to be 10 mol. % Cu₂Cr₄Te₇. After cooling in ice water, the glass transition region of the AsS-based system is 15 mol. % Cu₂Cr₄Te₇. The dependence of the softening temperature T_g, crystallization T_{crys}, microhardness and density of glassy region alloys on the composition was studied.

References:

1. Dinesh Chandra SATI1, Rajendra KUMAR, Ram Mohan MEHRA Influence of Thickness Oil Optical Properties of a: As₂Se₃ Thin Films // Turk J Phys. 2006. V.30. P.519- 527.
2. Lovu M., Shutov S., Rebeja S., Colomeyco E., Popescu M. Effect of metal additives on photodarkening kinetics in amorphous As₂Se₃ films // Journal of Optoelectronics and Advanced Materials 2000. V. 2. Issue: 1. P 53-58/
3. Jun J. Li Drabold, D. A. Atomistic comparison between stoichiometric and nonstoichiometric glasses: The cases of As₂Se₃ and As₄Se₄ // Phys. Rev. 2001. V. 64. P. 104206-104213.
4. Hineva T., Petkova T., Popov C., Pektov P., Reithmaier J. P., Funrmann-Lieker T., Axente E., Sima F., Mihailescu C. N., Socol G., Mihailescu I. N. Optical study of thin (As₂Se₃)_{1-x}(AgI)_x films // Journal of optoelektronics and Advanced Materials. 2007.Vol.9. No. 2. February. P. 326-329.
5. Seema Kandpal, Kushwaha R. P. S. Photoacoustic spectroscopy of thin films of As₂S₃, As₂Se₃ and GeSe₂ // Indian Academy of Sciences. PRAM ANA journal of physics. 2007. Vol. 69. No. 3 P. 481-484.
6. Littler I. C. M., Fu L. B., Mägi E. C., Pudo D., Eggleton B. J. Widely tunable, acoustooptic resonances in Chalcogenide As₂Se₃ fiber // Optics Express. 2006.V. 14. Issue 18. P. 8088- 8095.
7. Babaev A. A., Muradov R., Sultanov S. B., Askhabov A. M. Influence of production conditions on the optical and photoluminescent properties of glassy

As₂S₃ // Inorganic materials. 2008. Vol. 44. No. 11. P. 1187-1201.

8. Kim B.Y., Blake J.N., Engan H.E., Shaw H.J. All-fiber acousto-optic frequency shifter // Opt. Lett. 1986. V. 11. P. 389-391.

9. Lee S.S., Kim H.S., Hwang I.K., Yun S.H. Highly-efficient broadband acoustic transducer for all-fiber acousto-optic devices // Electron. Lett. 2003. V. 39. P. 1309-1310.

10. Engan H.E. Acousto-optic coupling in optical Fibers // IEEE Ultrasonics Symposium. 2000. V.1. P. 625-629.

11. Diez A., Birks T.A., Reeves W.H., Mangan B.J., St P. Russell J. Excitation of cladding modes in photonic crystal fibers by flexural acoustic waves // Optics Lett. 2000. V. 25. P. 1499-1501.

12. Claytor T.N., Sladek R.J. Ultrasonic velocities in amorphous As₂S₃ and As₂Se₃ between 1.5 and 296 K. // Phys. Rev. 1978. B. 18. P. 5842-5850.

13. Belov K.P., Tretyakov Yu.D., Gordeev I.V., Koroleva L.I., Pedko A.V., Bagorova S.D., Alferov V.A., Saksonov Yu.G., Shalimova M.A. Magnetic properties of chalcogenide spinels Cd_xCu_xCr₂S₄, CdCr₂Se_{4-y} FeCr₂S_{4-y} // FTT 1973. Vol. 15, No. 10. P. 3106-108.

14. North G.N. Anomalous photomagnetolectric effect in the ferrimagnetic semiconductor CdCr₂Se₄ // FTP. 1983. T. 17. No. 8. P.1505-507.

15. Shumilkin N. S. Interaction in Cu-In-Cr-Se(Te) Systems in the Region of Existence of Magnetic Phases with High Temperatures of Magnetic Ordering (Tc). Dis. for Cand. Sci. (Chem.). RAS, N.S. Kurnakov Institute of General and Inorganic Chemistry. 2002. 121 p.

16. Aliev I. I., Ismailova S. Sh., Guseynova Sh. A., Akhmedova D. A. Phase Equilibria in the As₂Te₃-Cu₂Cr₄Te₇ System Eurasian Union of Scientists 2021. 6/67. No. 1. pp. 14-17. DOI: 10.31618/ESU.2413-9335.2021.1.87.1389

17. Ismayilova S.Sh., Aliyev I.I., Ahmedova C.A., Alverdiyeva F.H., Gurbanov G.B. Study of glass formation in the As₂S₃-Cu₂Cr₄Te₇ system // Ganja State University International Scientific Conference. Actual problems of modern natural and economic sciences. Ganja. 2021. May 6-7. P.3-9.

18. . Khvorestenko A.S. Arsenic chalcogenides. Review from the series "Physical and chemical properties of solids". - Moscow, 1972. 92 p.